Novel Composite Materials for SOFC Cathode-Interconnect Contact

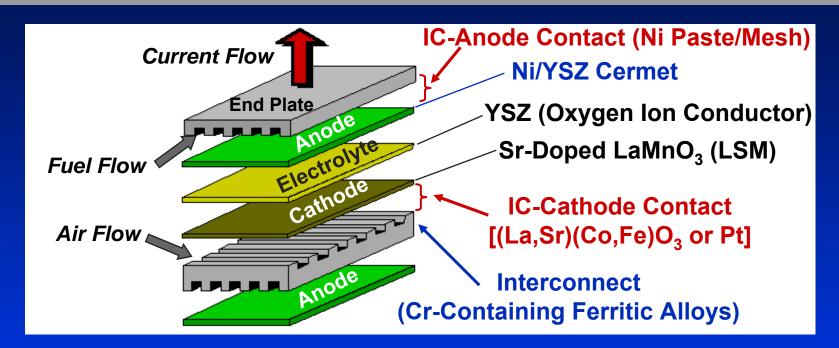
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Electrical Contacts in SOFC stacks



- Electrical contact layers are used to reduce the electrode/ interconnect interfacial resistance by compensating for the corrugations present on their respective surfaces
- Ni-paste in combination with Ni-mesh is widely used to establish electrical contact between interconnect and Ni+YSZ anode.
- Finding a suitable material for electrical contact between the cathode and interconnect is challenging

Criteria for Cathode/Interconnect Contact Materials

- Sufficiently high electrical conductivity over SOFC lifetime
- Reasonable match in coefficient of thermal expansion (CTE) with other cell components
- Chemical stability under high current condition
- Stability with both the interconnect and cathode under oxidizing environment, especially negligible effects on the formation of protective oxides on interconnect alloy
- High sinterability at the SOFC operating temperatures
- Ability to buffer the interactions between interconnect oxide (usually Cr-containing) and electrode oxide
- Tolerance to thermal cycle-induced damage (possible self-healing if thermal cycle-induced cracking occurs in the contact layer)

Objectives of this Research Effort

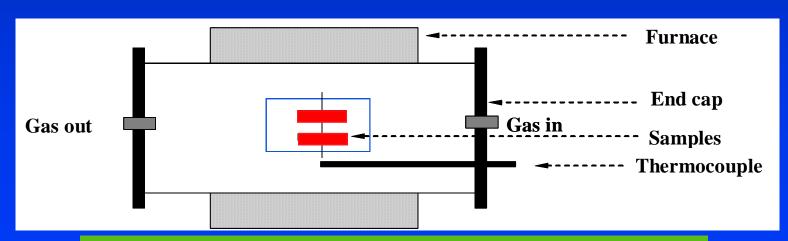
The overall goal of this project is to develop Agperovskite composites with reduced Ag evaporation, ability to absorb Cr from the interconnect, and adequate damage-tolerance as interconnect-cathode contact.

- Elucidation of the mechanism of Ag evaporation at elevated temperatures
- Alloy design of new Ag alloys with reduced Ag evaporation/migration
- Optimization of processing and microstructures of the composite contact of Ag + perovskite
 [(La_{0.6}Sr_{0.4})(Co_{0.8}Fe_{0.2})O₃ LSCF or (La_{0.8}Sr_{0.2})MnO₃ LSM]
- Demonstration/assessment of performance of the new contact materials

Ag Evaporation: Pure Ag, Ag-base Alloys, and Composites

Ag Evaporation Testing

- Preparation of Ag evaporation samples
 - Ag-base alloys were prepared by arc melting, drop casting, followed by repeated cold rolling and annealing
 - Ag-perovskite composites were prepared by powder metallurgy
- The effect of various low-cost alloying elements such as Ti, Fe, Co, Ni, Mn, Sn, and Zn at levels of 1, 5, and 15at.% as well as noble metals such as Pd on the Ag evaporation rate were determined.
- The Ag evaporation behavior of the Ag+LSM and Ag+LSCF composites with different Ag-to-perovskite ratios was also evaluated.



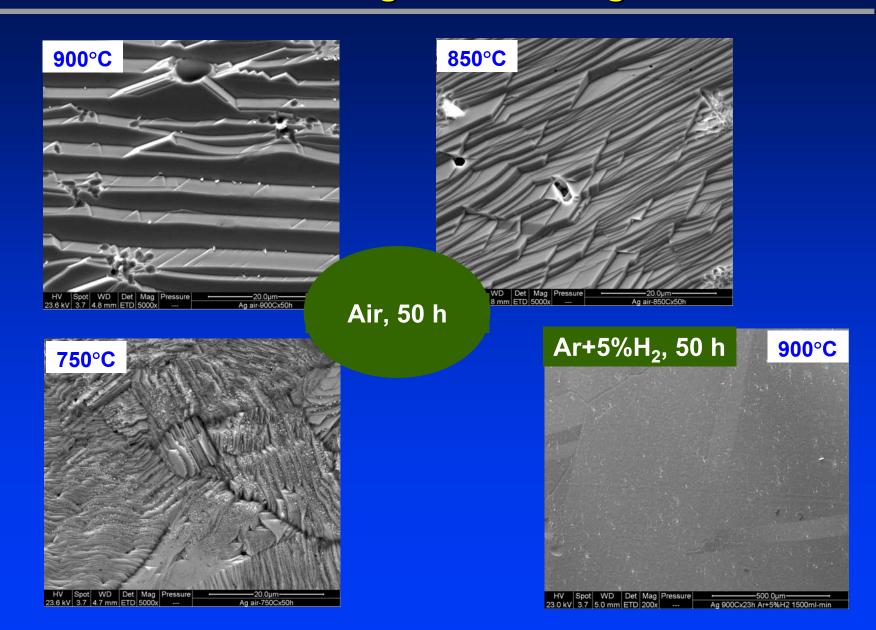
Experimental Setup for Ag Evaporation Testing

Effect of Atmosphere on Ag Evaporation

Atmosphere	Flow Rate (cm.s ⁻¹)	Temperature (°C)	Evaporation Rate (x10 ⁻⁹ g.cm ⁻² .s ⁻¹)
Ar+5%H ₂ +3%H ₂ O	1.5	900	7.12
Ar+5%H ₂	1.5	900	7.06
Air+3%H ₂ O	1.5	900	7.22
Air	1.5	900	7.11

- The exposure atmospheres showed no measurable influence on the evaporation rate of Ag at 900°C.
- The essentially same evaporation rate in different environments indicates that the exposure atmosphere is not involved in the critical step that controls the Ag evaporation kinetics.
- The effect of atmosphere on Ag evaporation at lower temperatures are currently being evaluated.

Different surface morphologies were observed after evaporation test in oxidizing and reducing environments

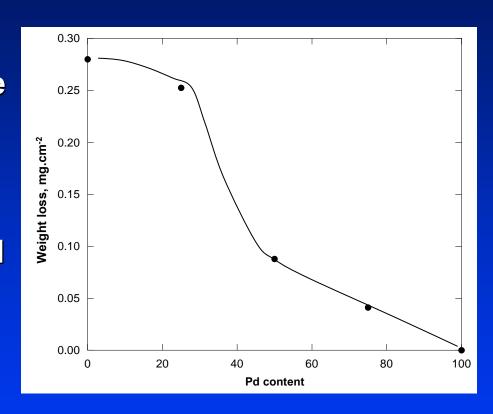


Plausibility of Ag as Cathode-Interconnect Contact

- At 800°C and air flow rate of 1.1 cm/s, the evaporation rate of pure Ag is 4x10⁻¹⁰ g.cm⁻².s⁻¹
- Considering the SOFC target lifetime of 40,000 h, this value is too high (0.06 g.cm⁻² or 60 µm)
- Since Ag evaporation is determined by the bonding of Ag atoms, alloy design should focus on identifying:
 - Alloying elements that drastically affect the bonding in Ag
 - Surface-active elements that block the surface Ag sites

Effect of Noble Metal Additions on Ag Evaporation

- With the addition of 25%Pt, Au, and Pd, some reduction in Ag evaporation rate in the range of 10-30% was observed for all the alloying elements.
- As the Pd content increased in the alloys, the Ag evaporation rate dropped consistently.
- Additions of the noble metals (Pt, Au, and Pd) were not practical due to high cost of the raw materials



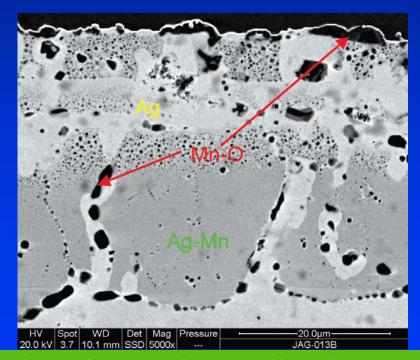
Weight losses of various Ag-Pd alloys after thermal exposure at 850°C for 40 hours in air with a flow rate of 1.5 cm.s⁻¹

Effect of Low-Cost Metal Addition on Ag Evaporation

- A significant reduction in weight loss was observed with the Mn and Zn additions into Ag after an initial exposure time; during subsequent exposures, however, a relatively constant evaporation rate similar to pure Ag was observed.
- This behavior was a result of selective oxidation of the alloying element, which led to the initial weight gain; furthermore, the thermally-grown oxide scale was highly porous, not effective in blocking Ag from evaporation.
- Alloying with small amounts of low-cost elements had no beneficial effect on the long-term Ag evaporation rate.

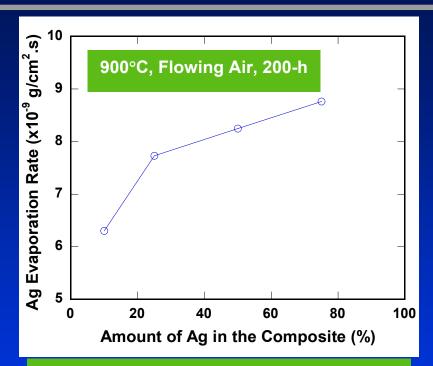
Ag Evaporation Rate at 800°C During 100-h Air Exposure for Ag-5%Mn

100-h Exposures	1 st	2 nd	3 rd
Evap. Rate (g/cm².s)	-6x10 ⁻⁹	8x10 ⁻¹⁰	7.5x10 ⁻¹⁰

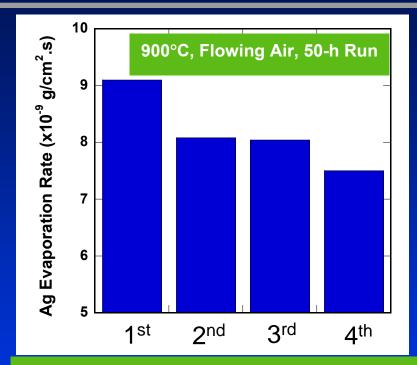


Cross Section of a Ag-15%Mn Coupon after 160-h Exposure to Air at 900°C

Ag Evaporation: Ag-Perovskite Composites



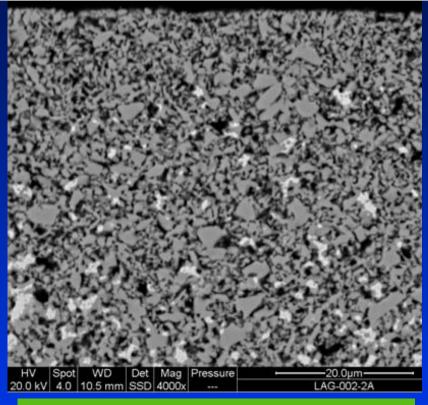
Effect of the Volume % of Ag in the Composite on the Evaporation Rate



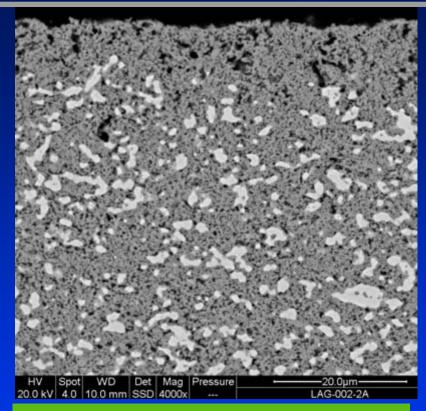
Ag Evaporation Rate of Ag-50%LSM as a Function of Exposure Time (Run)

- Composite materials of Ag+LSM exhibited a slightly reduced Ag evaporation rate as the amount of the perovskite phase increased. Even with 90%LSM, the evaporation rate was still pretty high, which can be attributed to the porous structure of the composite material.
- As the evaporation time increased, the Ag evaporation rate decreased slightly for the composite materials.

Ag Evaporation: Ag-Perovskite Composites



Cross-sectional View of Ag+LSM (25%Ag) after air exposure for 200 h



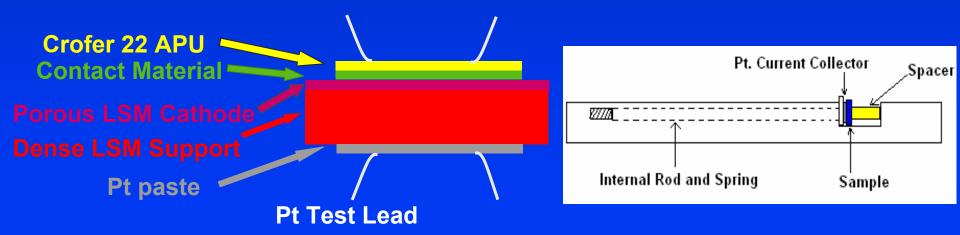
Cross-sectional View of Ag+LSCF (25%Ag) after air exposure for 100 h

- A Ag-free zone near the surface was observed for both the Ag+LSM and Ag+LSCF composite, due to the porous structures.
- More extensive Ag depletion was observed for more porous structure and longer exposure.

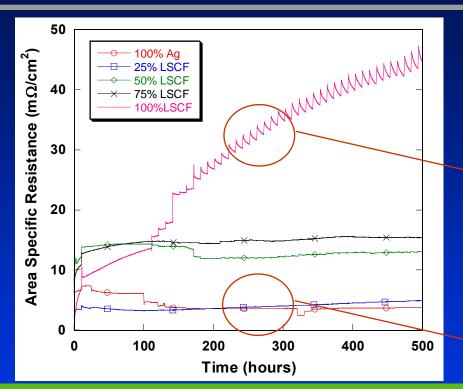
Performance of Ag-Perovskite Composites as SOFC Contact Materials

Evaluation of the Ag+Perovskite Composites

- A number of the Ag+LSM and Ag+LSCF composites with the Ag content of 0, 10, 25, 50, 75, and 100% (in volume) in the composite have been processed and interconnect/contact/cathode test cells with the Ag-perovskite composites as contact material have been constructed.
- The test cells were placed in the spring-loaded alumina rig to apply a small compressive load throughout testing, of ~0.16 kg/cm²; a constant current density of 250 mA/cm² was used.
- The change in area-specific resistance (ASR) was recorded as a function of exposure time during isothermal or cyclic exposure.

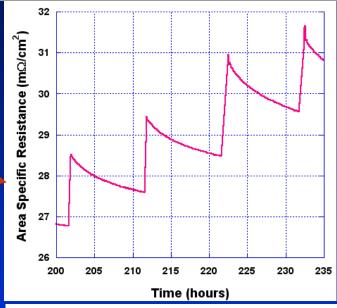


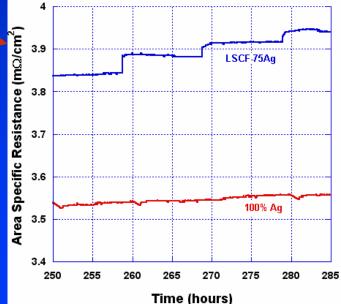
As Ag content increased in the Ag+LSCF composite, the ASR decreased correspondingly



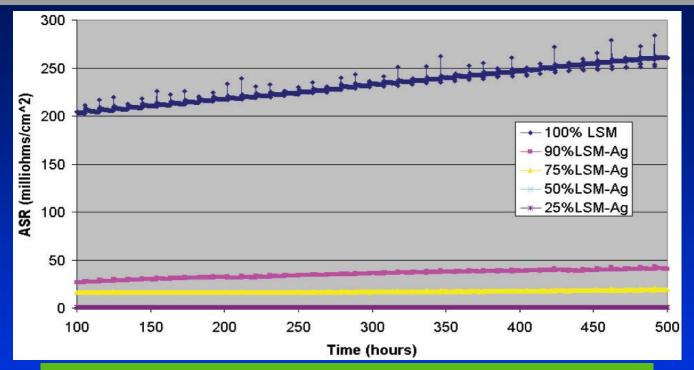
ASR change during thermal cycling (initial firing of 850°Cx10h; "break-in" of 800°Cx100h; 40 cycles with 800°Cx10h holding plus furnace cooling to 250°C)

- Less damage was observed during thermal cycling for Ag-containing contact materials
- Some "self-healing" was observed for LSCF





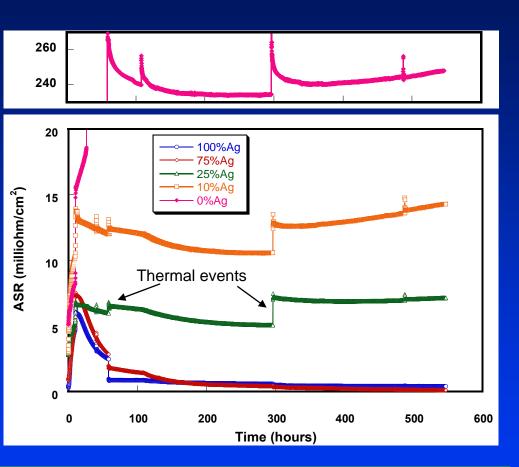
Similar results were obtained for the Ag+LSM composites during thermal cycling



ASR change during thermal cycling (50 cycles with 800°Cx10h holding plus furnace cooling to 250°C)

- The pure LSM contact gave much higher ASR than pure LSCF; the addition of even 10% Ag into LSM drastically reduced the overall ASR
- Based on the projected values of the ASRs after 40,000 h using their rate of increase as determined from the figure, at least 25% Ag is needed to achieve the expected lifetime of 40,000 h and <100 m Ω /cm².

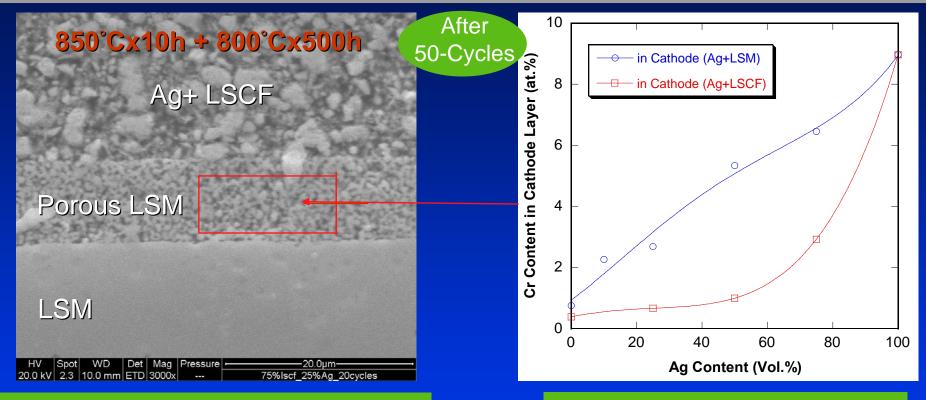
ASR Change of the Ag+LSM Composites during Isothermal Exposure



ASR Change during Isothermal Exposure with Occasional Cycling (Furnace Cooling to 250°C)

- exhibited much higher ASR than other compositions; the addition of even 10% Ag into LSM drastically reduced the overall ASR
- The isothermally-exposed test cells exhibited ASR degradation rates that were an order of magnitude lower than the thermally-cycled test cells of the same composition.
- The feasibility of using composite contact pastes with less than 25% Ag addition was confirmed.

With the increase of LSCF and LSM in the contact layer, Cr migration to the cathode was reduced

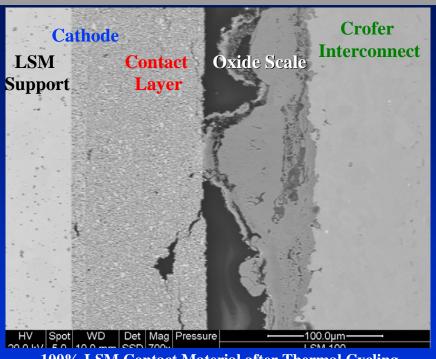


Cross-Sectional View of a Thermally-Cycled Cell

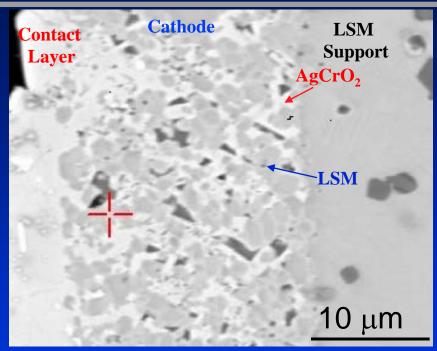
Cr Content in the Porous LSM Cathode

- A lower amount of Cr in the contact layer and a higher Cr content in the cathode was observed for the Ag+LSM contacts than for Ag+LSCF, implying LSM is not as effective as LSCF at trapping Cr within the contact layer.
- As the perovskite content in the contact increased, the Cr content in the porous LSM cathode decreased significantly.

SEM Cross-Sectional Observation of the Cycled Cells



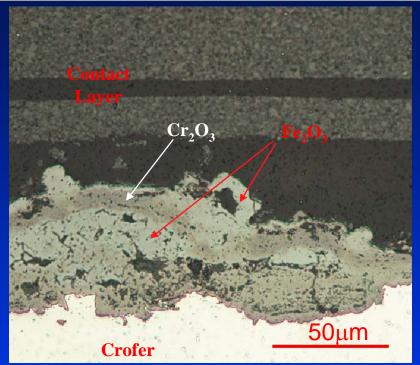
100% LSM Contact Material after Thermal Cycling

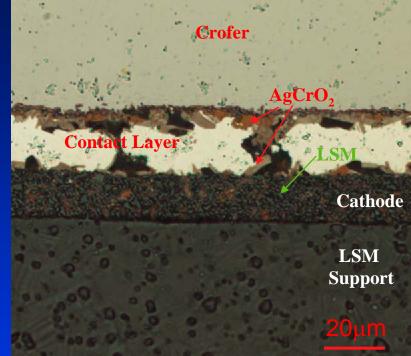


100% Ag Contact Material after Thermal Cycling

- The 100% LSM contact material proved to be very brittle in nature and separated from the interconnect upon removal from the test fixture.
- SEM analysis revealed the presence of a very thick layer of Fe₂O₃ on the interconnect for the 50-100% LSM contact material test cell
- Significant Ag migration into the cathode was observed for the Ag-containing contact materials, which led to the formation of (Ag,Mn)CrO₂.
- Similar results were observed for the cycled Ag+LSCF composites.

Similar results were observed for the Ag+LSM composites after isothermal exposure





100% LSM Contact Material after Isothermal Exposure

100% Ag Contact Material after Isothermal Exposure

- The presence of a very thick layer of Fe₂O₃ on the interconnect was confirmed for the 50-100% LSM contact cell after isothermal exposure with 2 cycles.
- Significant Ag migration into the cathode was observed for the Ag-containing contact materials, which led to the formation of (Ag,Mn)CrO₂.
- The interaction of Ag with the oxide scale thermally grown on the interconnect alloy also caused the formation of (Ag,Mn)CrO₂ at the interface.

Remaining Issues related to the Ag+Perovskite Composites

- Is the formation of the thick, non-protective oxide scale formed on the Crofer interconnect related to thermal cycling?
- What is the effect of current on the formation of such oxide scale?
- What is the effect of interconnect alloy composition and/or coating on the formation of such oxide scale?
- What is the minimum Ag content to achieve the adequate ASF during isothermal exposure?
- What is the role of Ag in reducing the ASR of the cell with the composite contact?
- Is there any more effective metal or oxide that can achieve similar results?

Acknowledgements

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